

Mechanical properties of carbon nanotube webs

ANAM
initiative

J. C. Stallard, W. Tan, N. A. Fleck,
A. M. Boies, F. R. Smail.

The ANAM Initiative.



Broad collaboration between **Cambridge Engineering** and **Materials Science** Departments, and **Ulster University**, and several **industrial partners**.

Focus on **direct-spun CNT materials**, made by the **Windle Process**



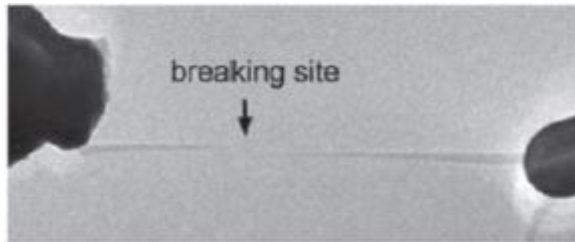
Carbon Nanotubes: Intrinsic Properties

Individual Tubes: MWNT
wall structural properties

$$E = 1 \text{ TPa}$$

$$\sigma_f > 100 \text{ GPa}$$

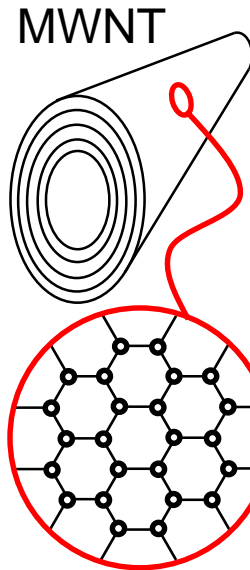
$$\rho \sim 2200 \text{ kg/cm}^3$$



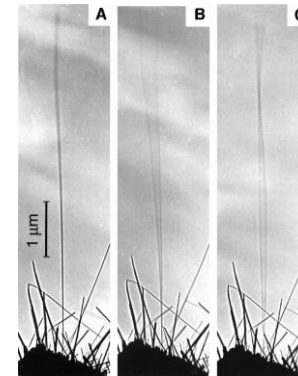
Wang *et al.*, (2010)

Good **understanding** of
mechanics with **strong**
theoretical validation

Yu *et al.* (2000)
Wang *et al.* (2010)
Zhang *et al.* (2014)



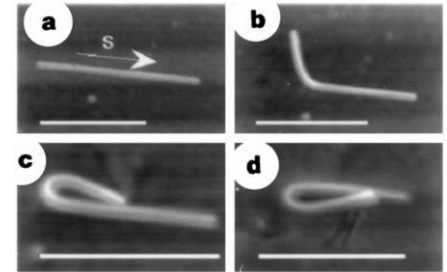
C=C double bond
Hexagonal Lattice



(Poncharal *et al.*, 1999)
Determination of modulus by electrostatic vibrations

(Falvo *et al.*, 1997)

Large elastic deformation

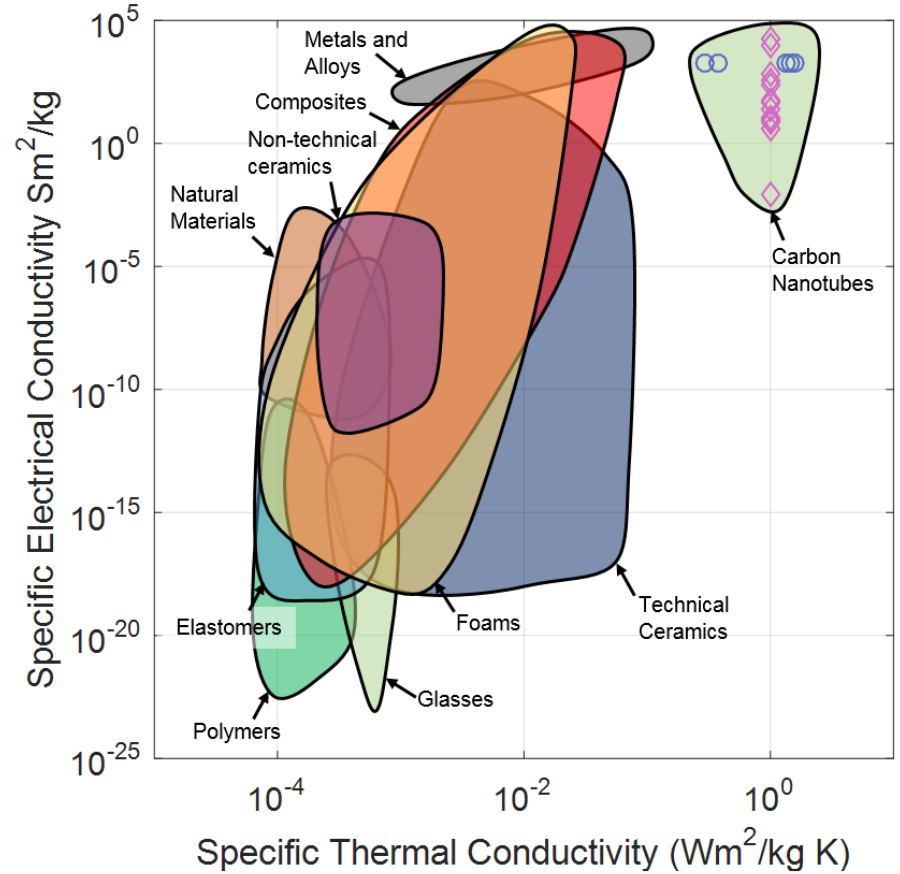
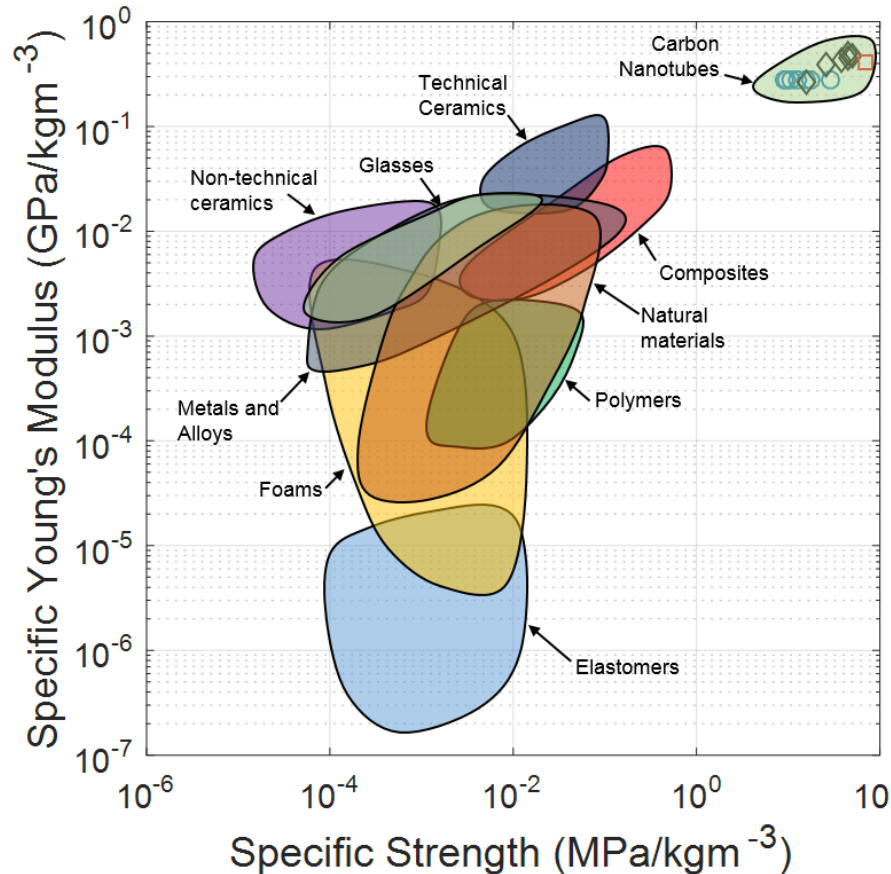


Electrical Conductivity: **$2 \times 10^5 \text{ s/cm}$**
Thermal Conductivity: **3500 W/mK**

- Can we realise the **properties of CNTs** in **Direct-spun Mats** and other **Bulk CNT Materials**?
- If not, **why**?

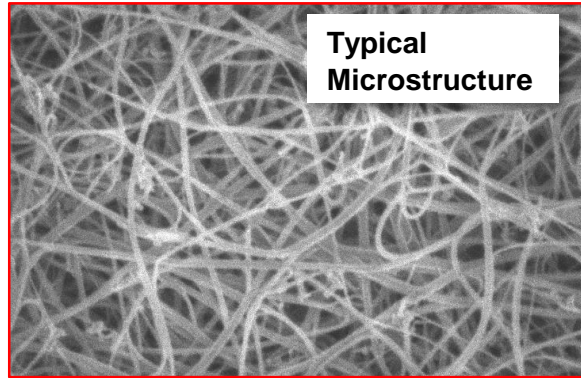


Carbon Nanotubes: Intrinsic Properties

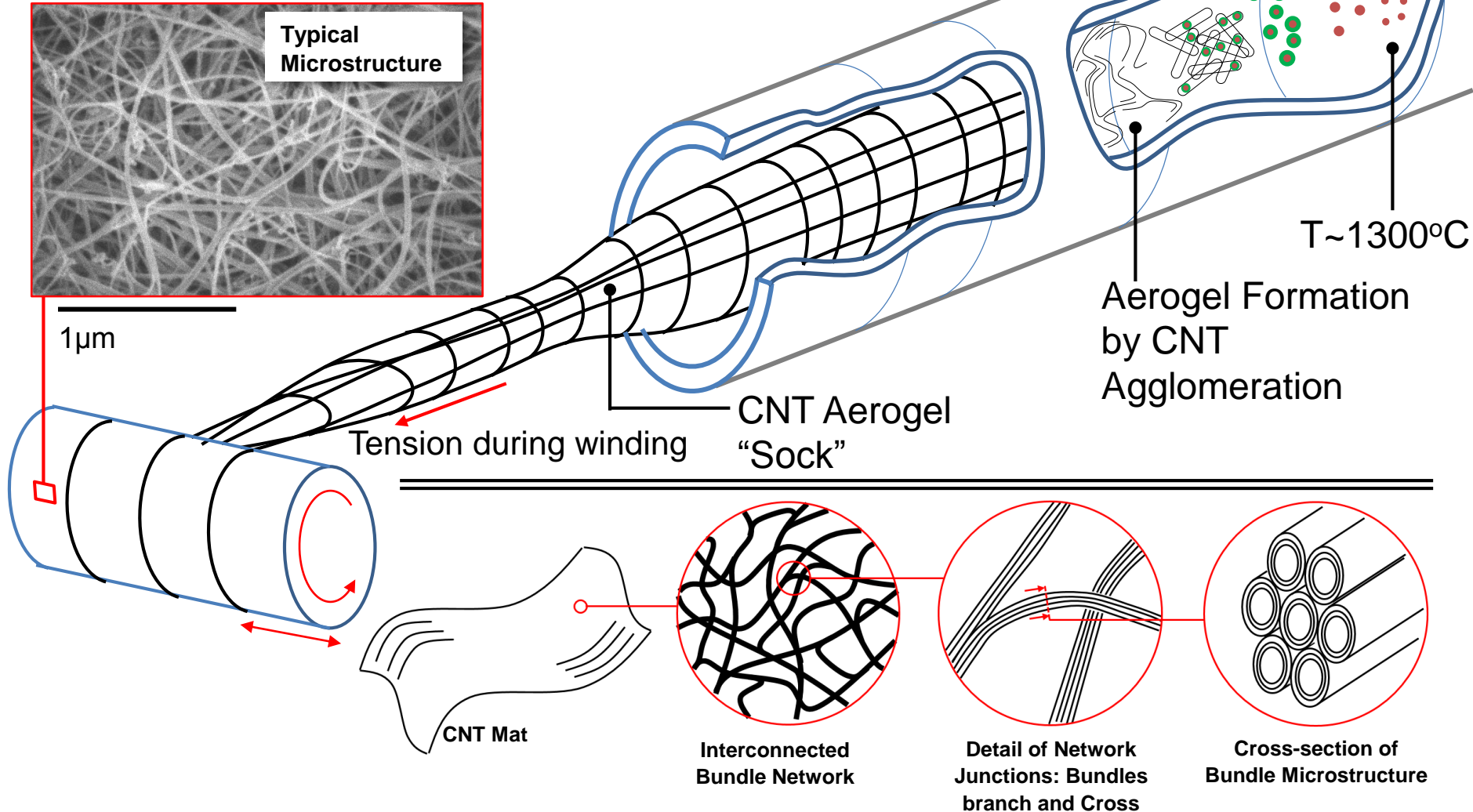


Production and Microstructure of direct-spun CNT Mat & Fibre from the 'Windle Process'

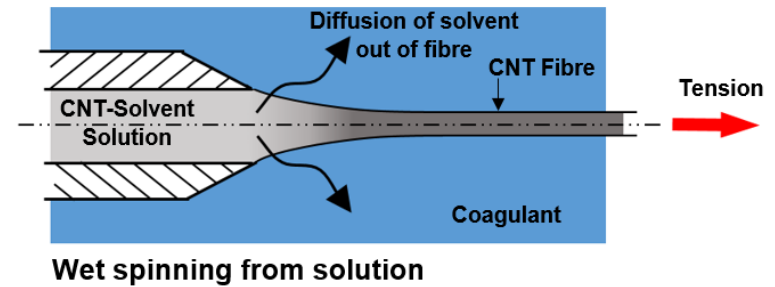
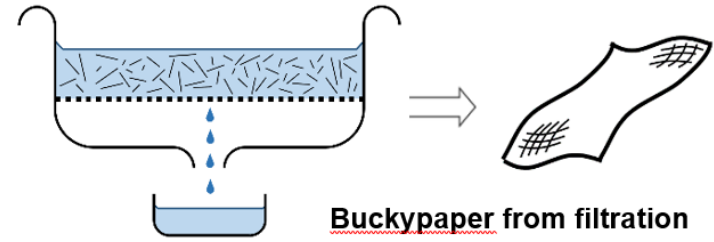
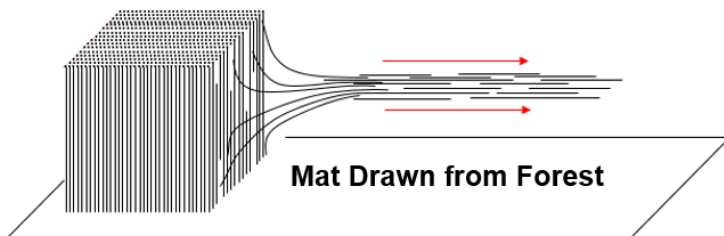
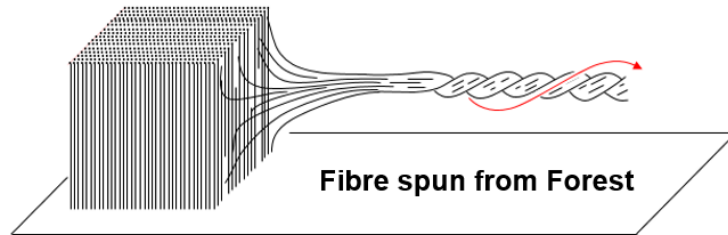
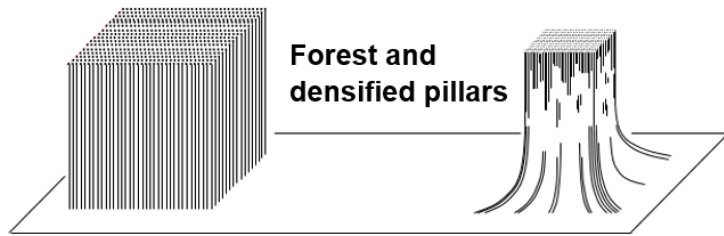
Methane, Ferrocene, Thiophene



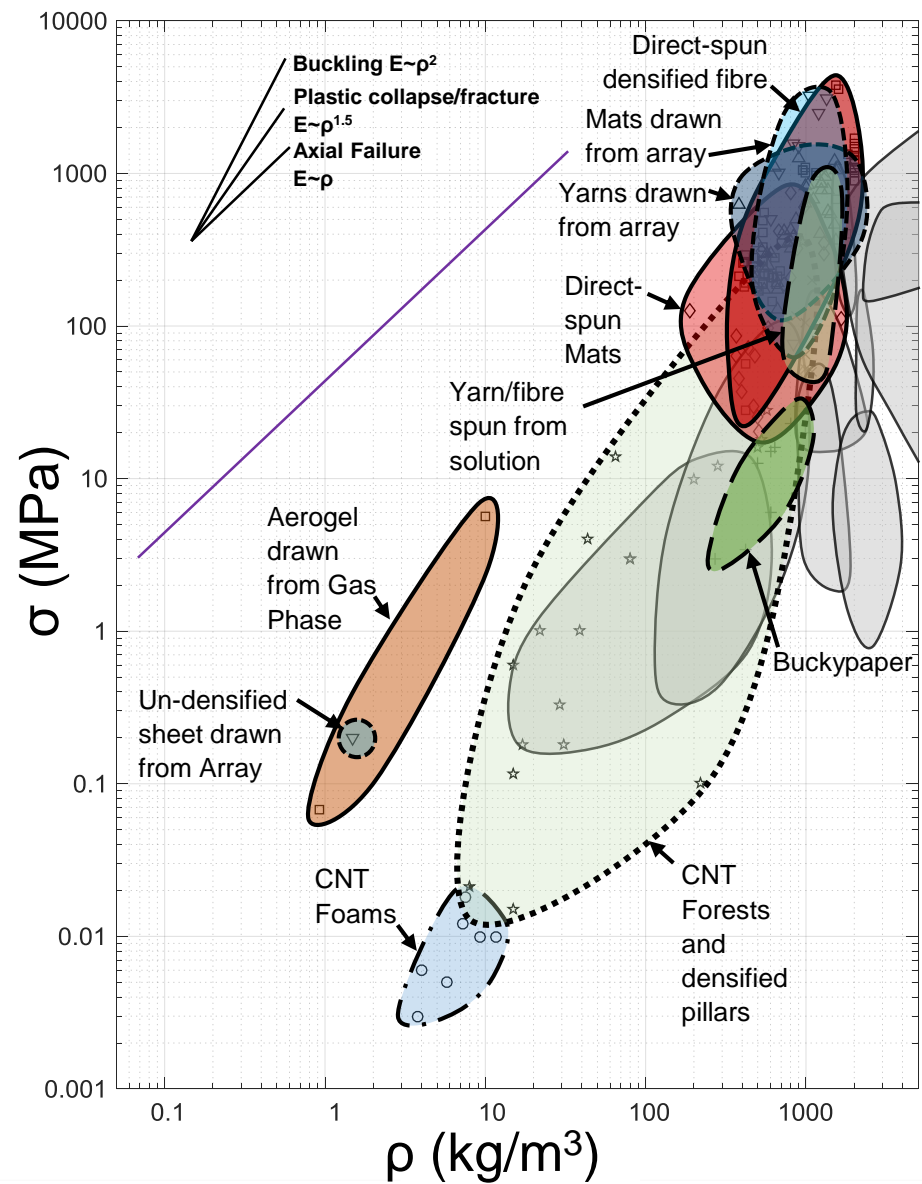
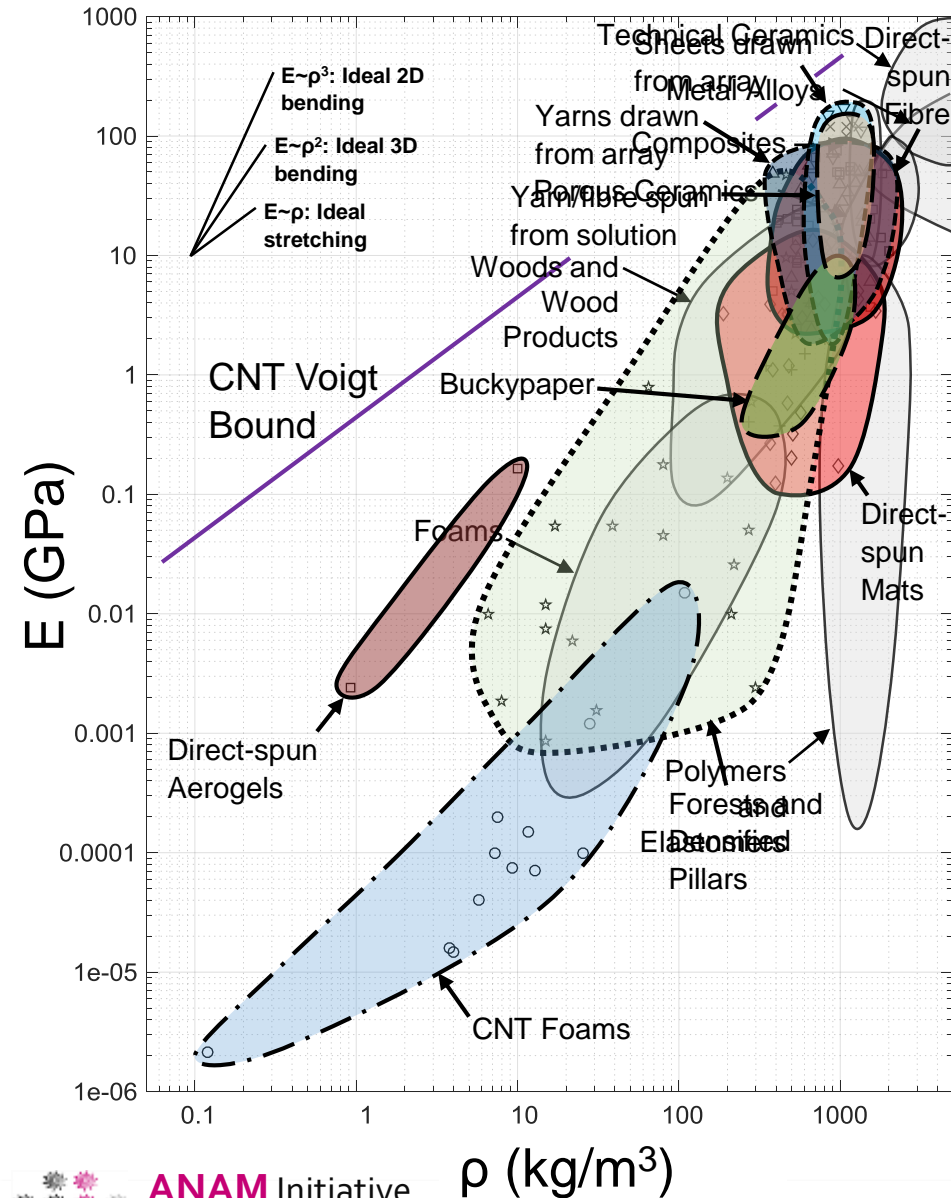
1µm



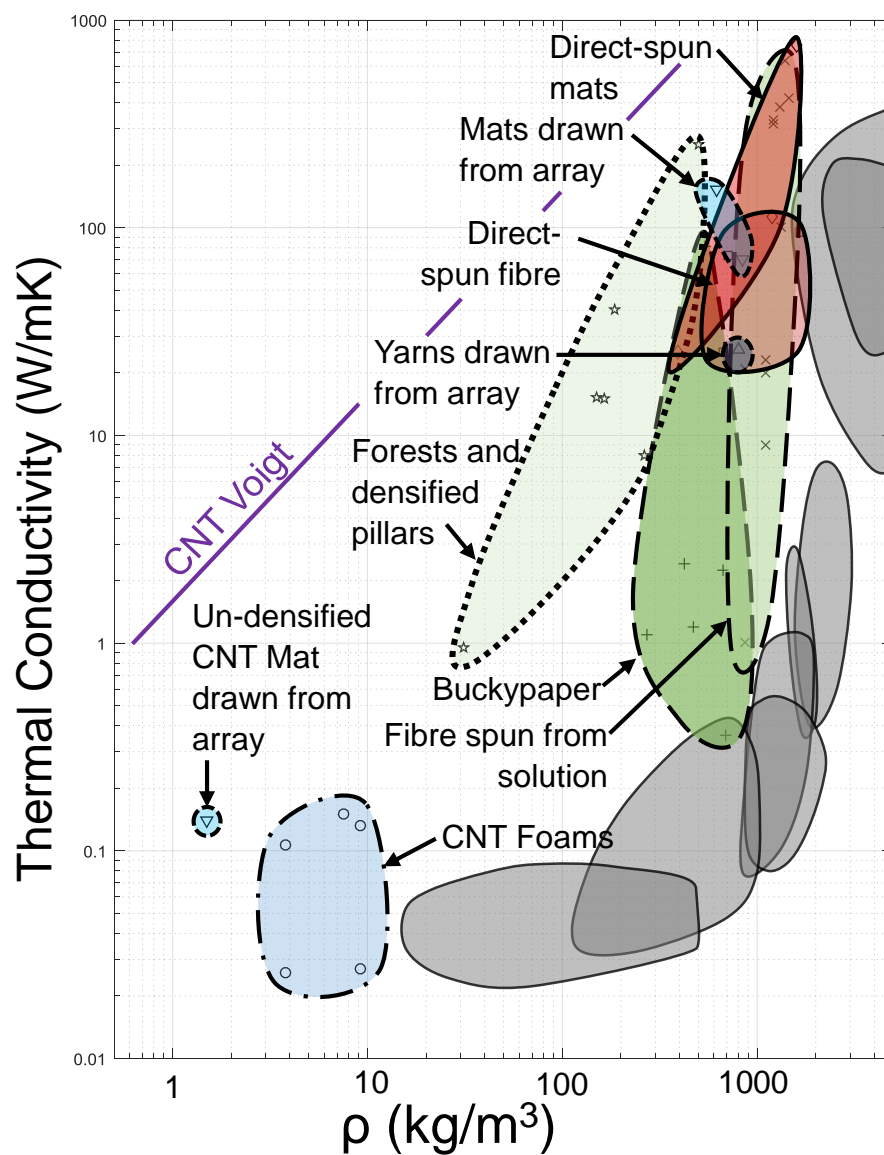
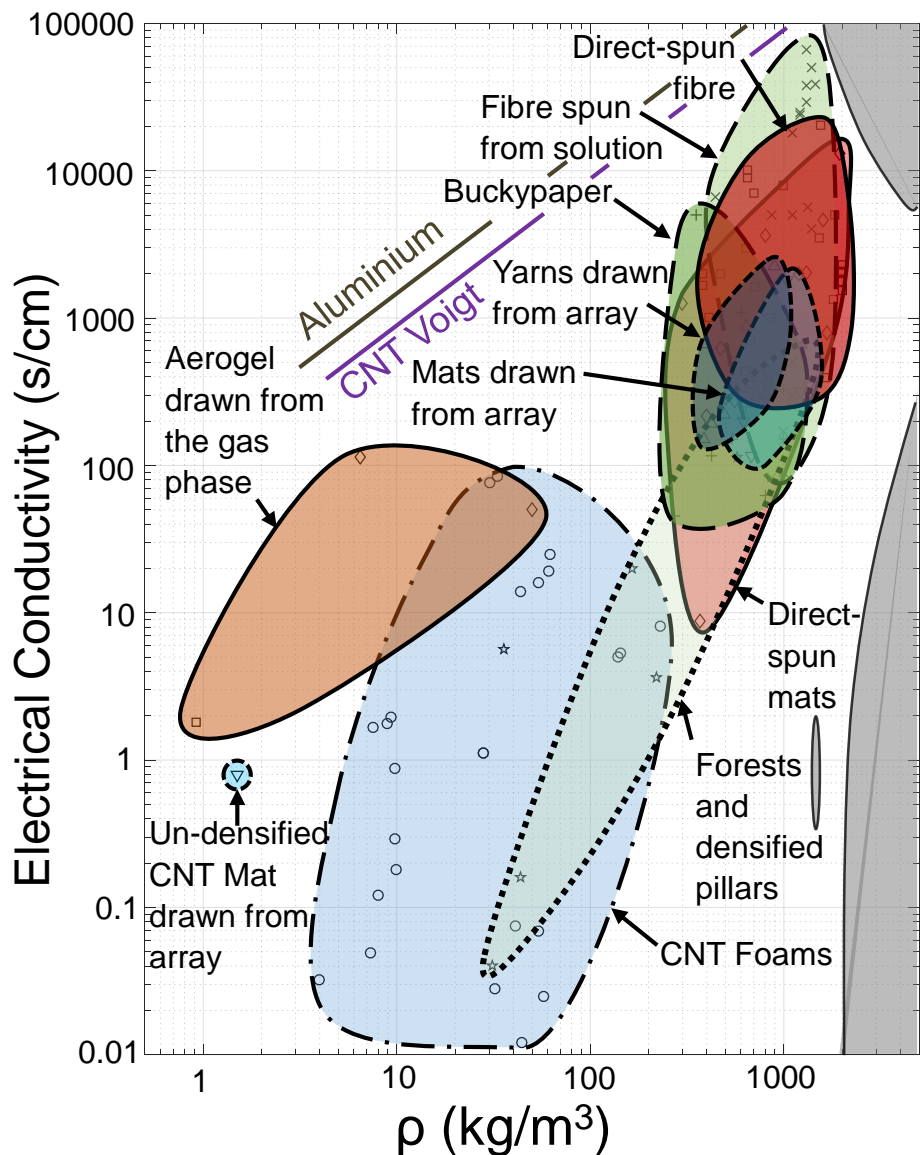
Bulk CNT Materials: *methods of manufacture*



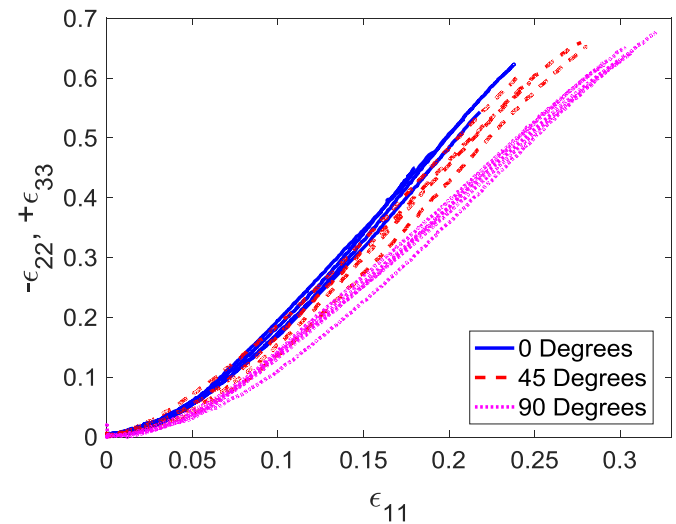
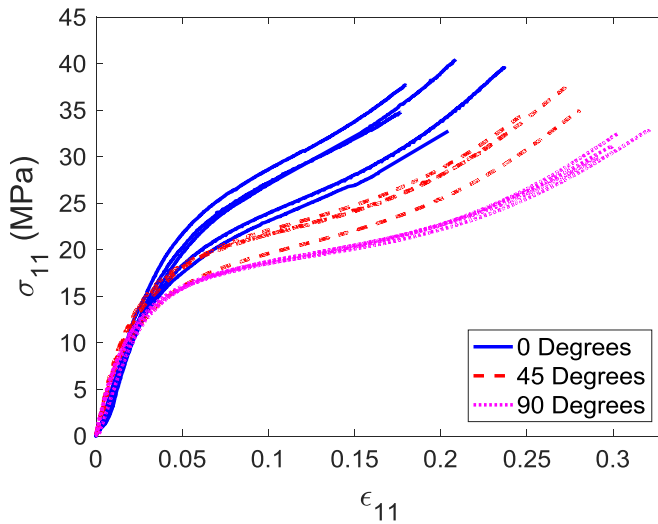
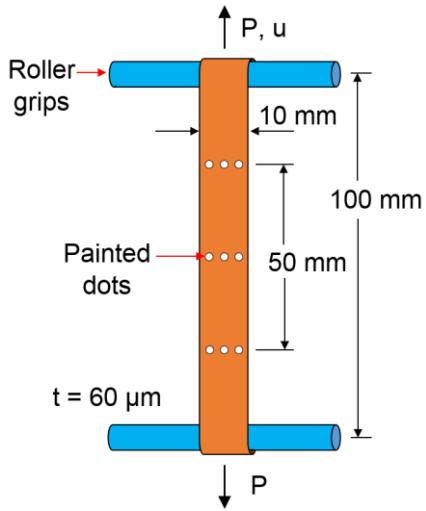
The Properties of Bulk CNT Materials: Mechanical



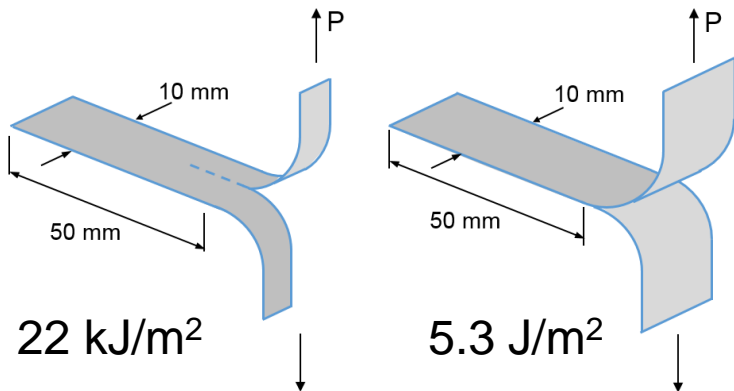
The Properties of Bulk CNT Materials: Electrical & Thermal



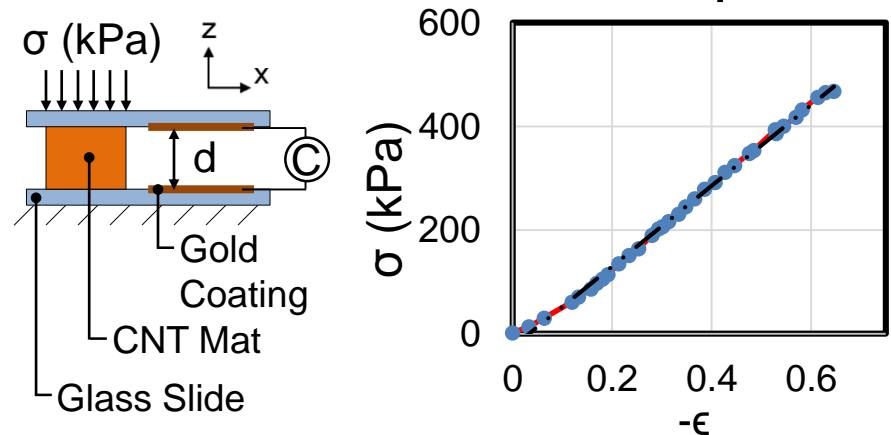
The Properties of a Direct-spun CNT Mat: Uniaxial Response, composition, and electrical properties



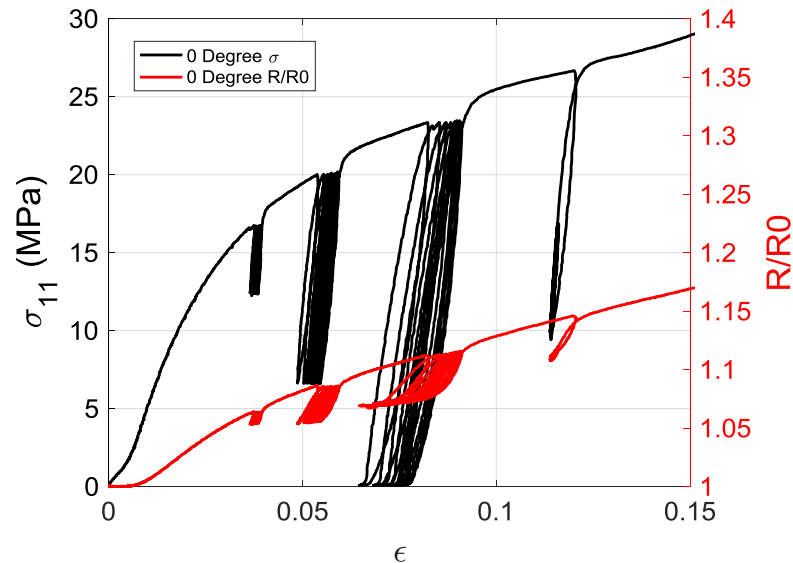
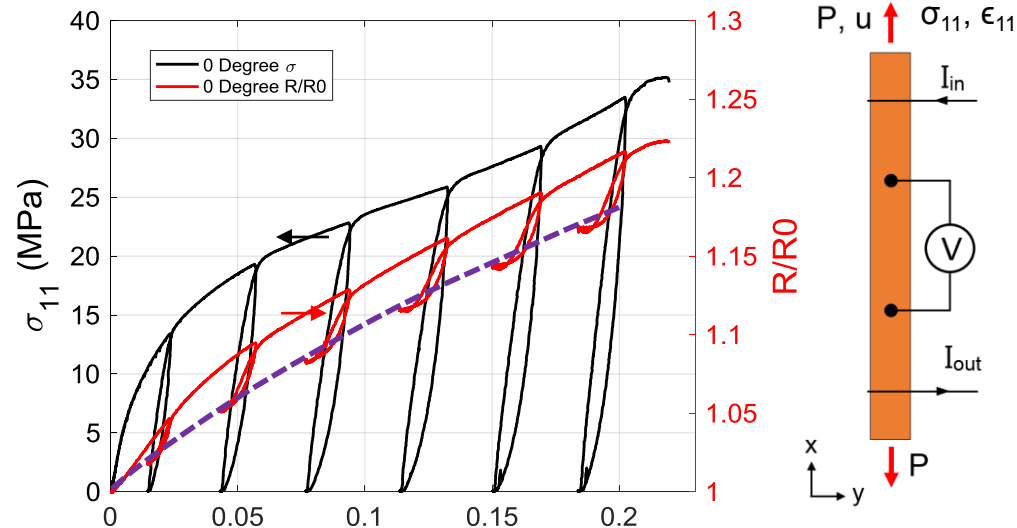
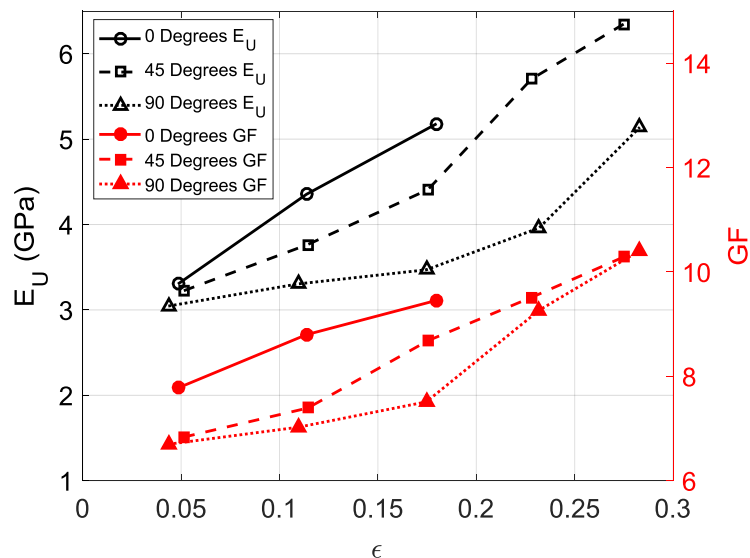
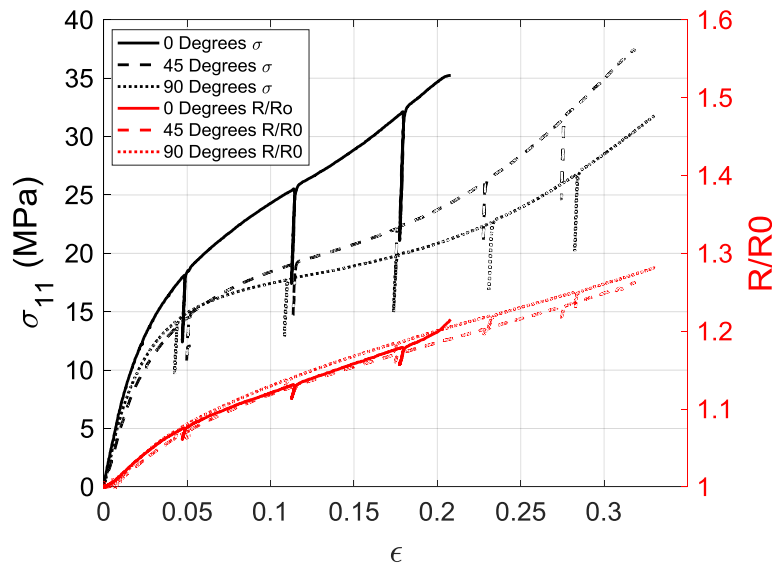
Fracture and Delamination



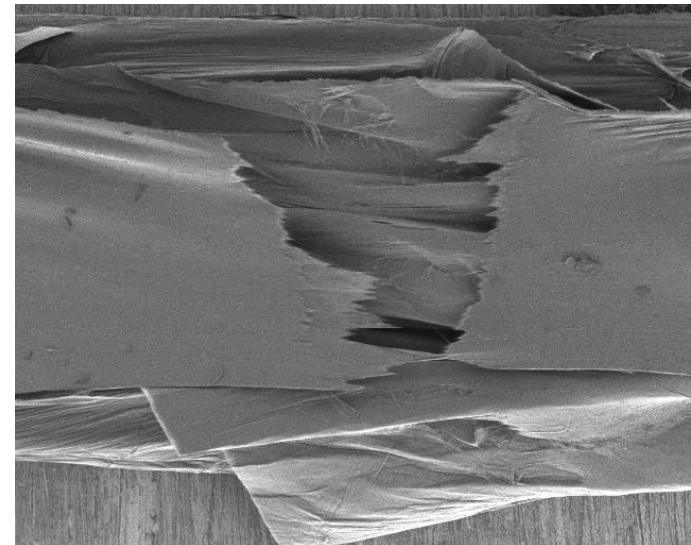
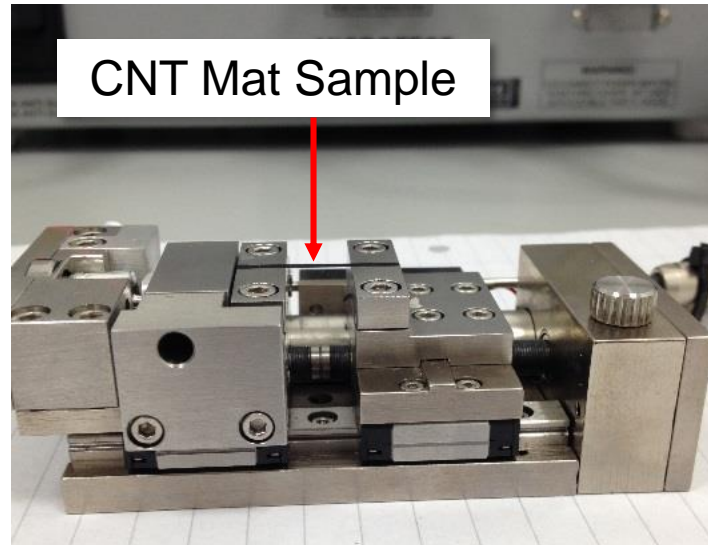
Out of Plane Response



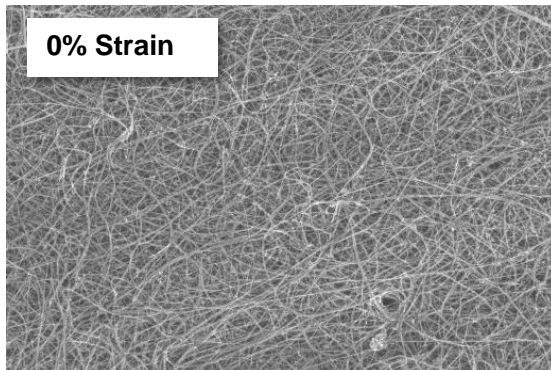
Direct-spun CNT Mat: In-Plane Piezoresistivity, and Unloading



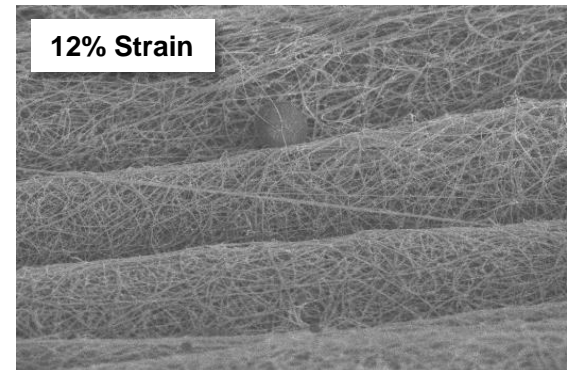
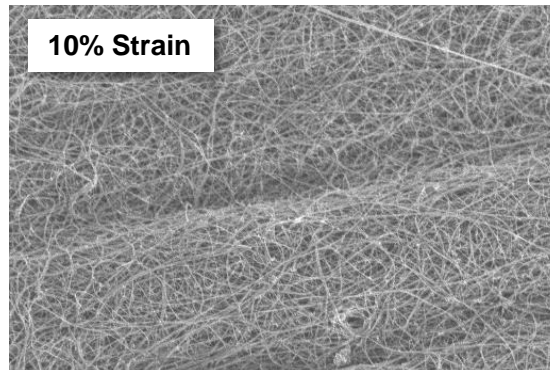
In-Situ Tensile Testing



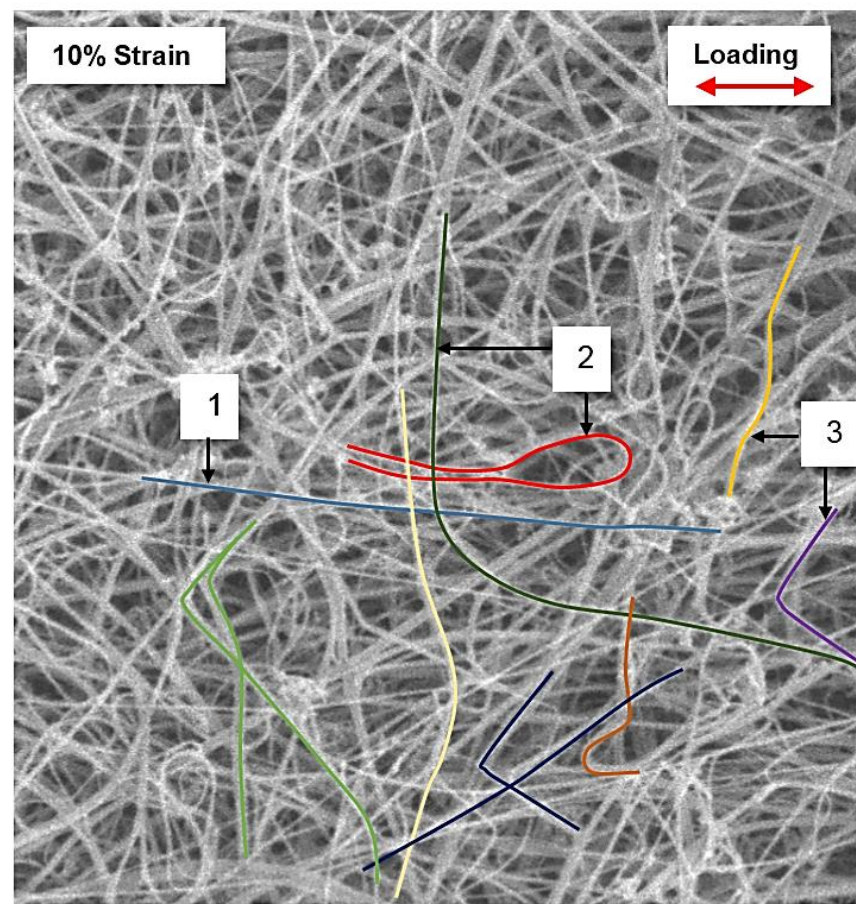
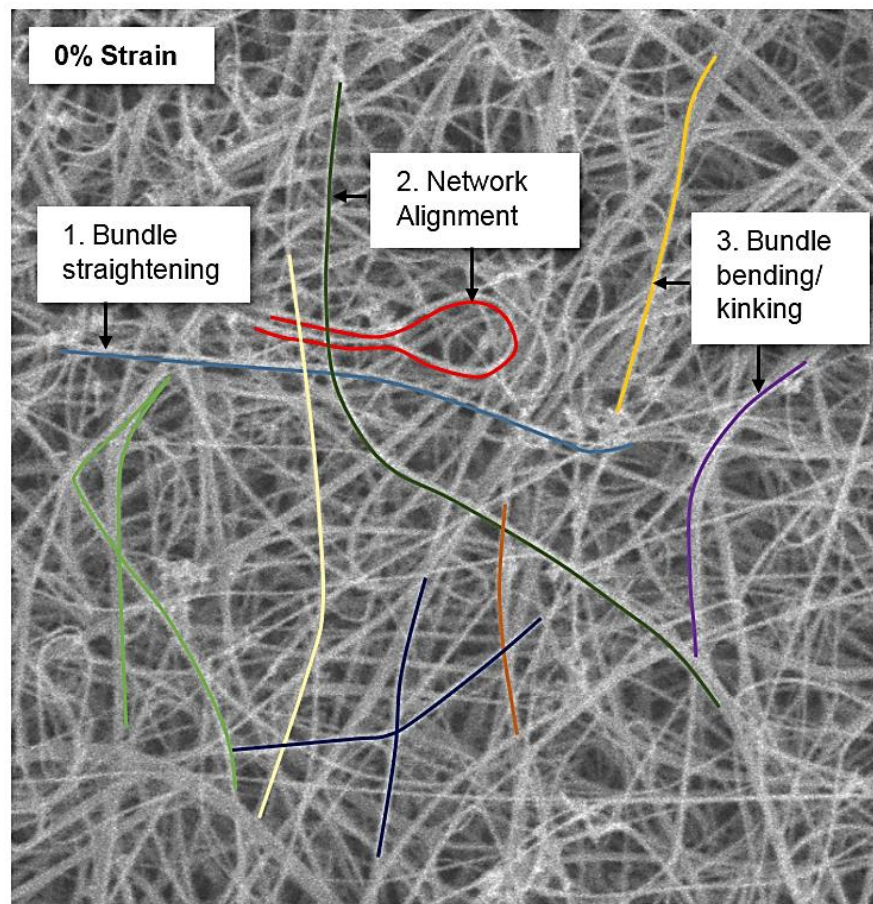
500 μm



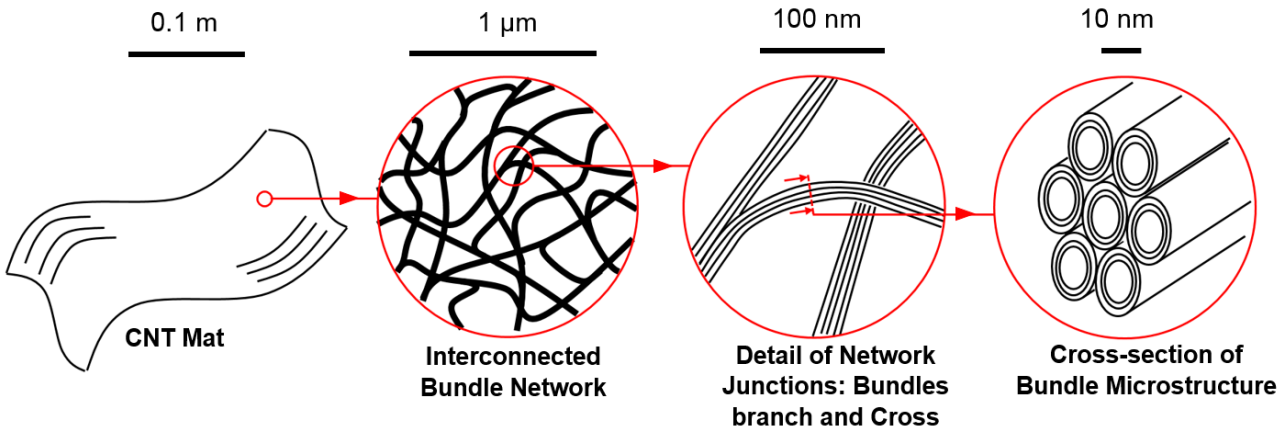
5 μm



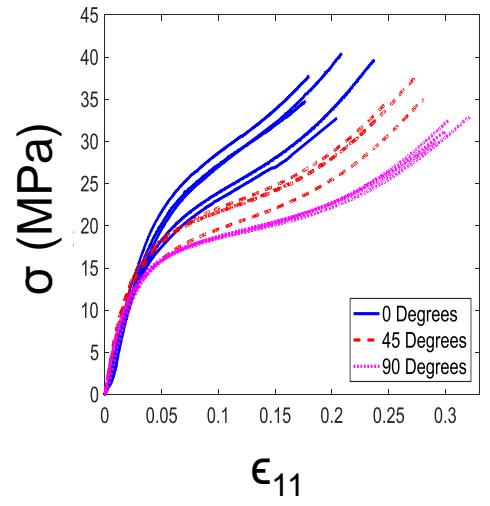
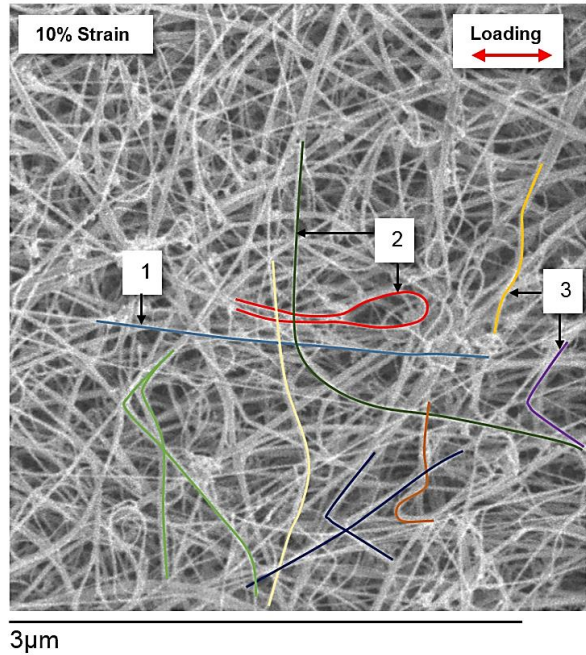
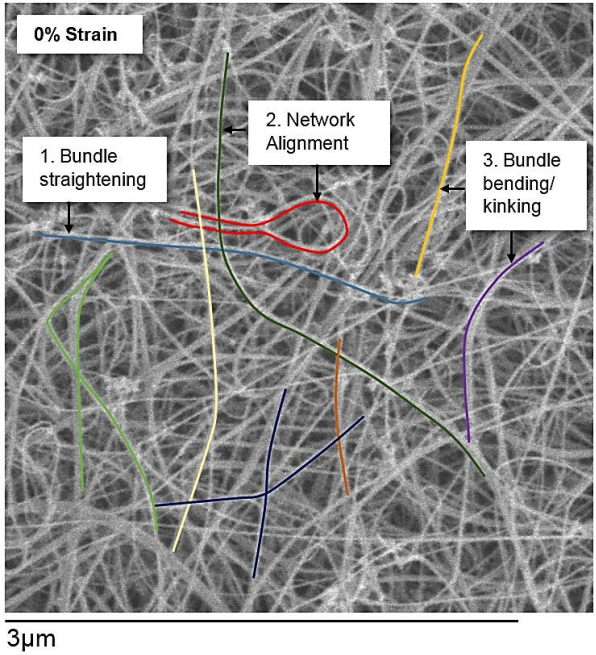
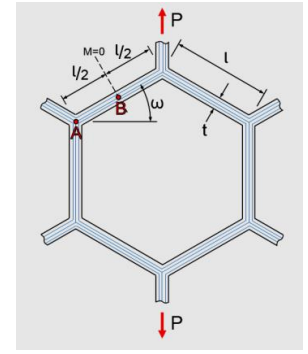
Microstructural change during the uniaxial response



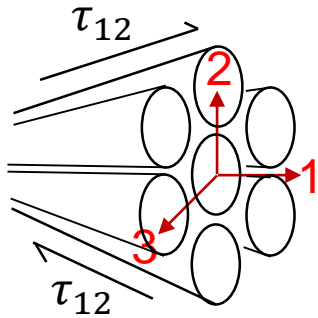
Direct-Spun Mats: the story



- **Rope-like** CNT bundles form random interlinked **bundle network**
- Network **deforms** like a **foam**, with **transverse deflection of struts**.

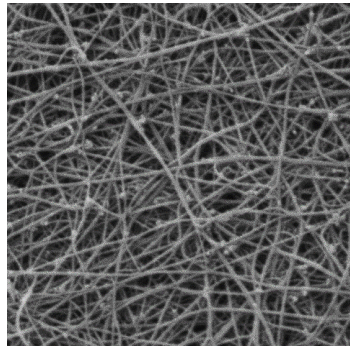


Micromechanical Model for direct-spun mat



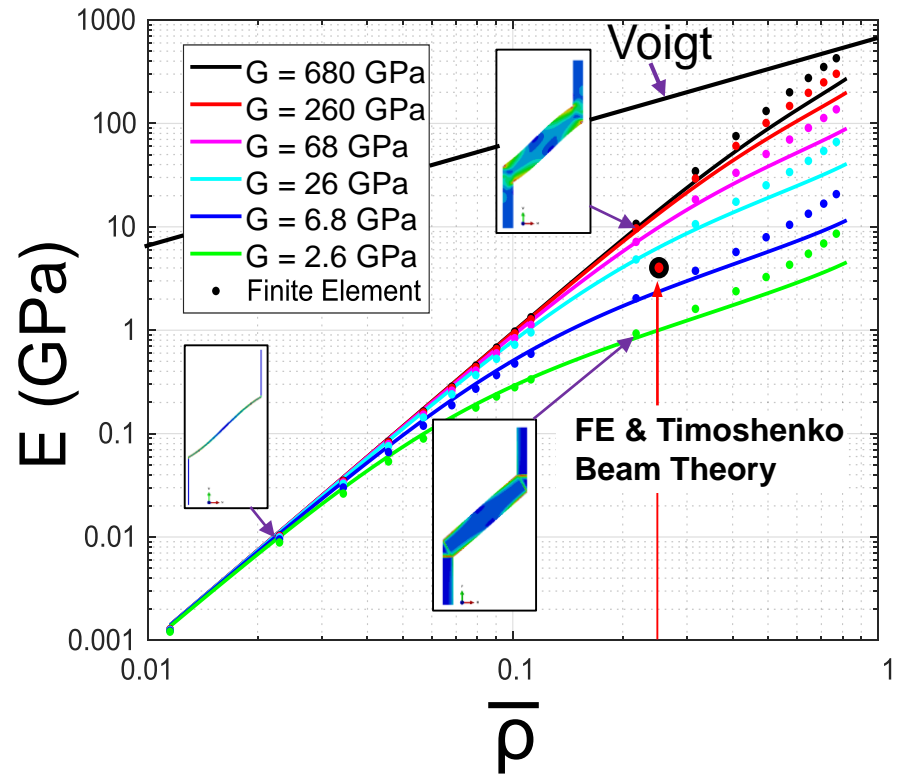
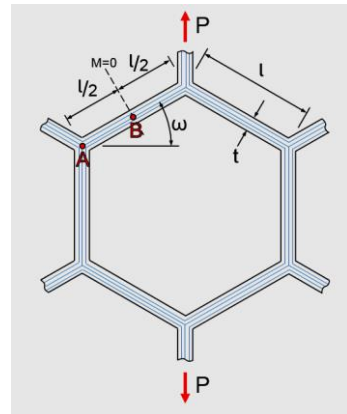
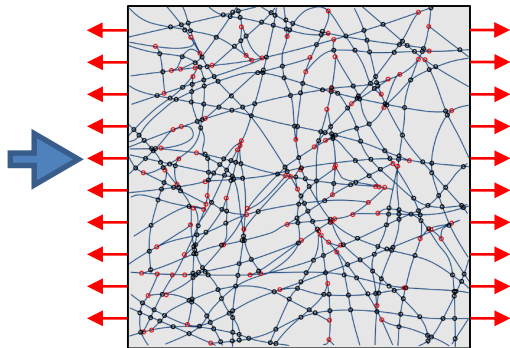
CNT Bundles are **Anisotropic**:

$E_{11} = 680 \text{ GPa}$
 $G_{12} = G_{23} = 9.5 \text{ GPa}$
 $E_{22} = E_{33} = 50 \text{ GPa}$
 $\nu_{12} = \nu_{13} = \nu_{23} = 0.3$



1 μm

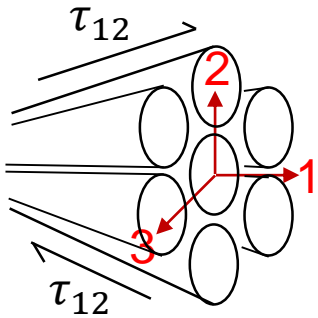
Approximate network with a **periodic honeycomb** unit cell



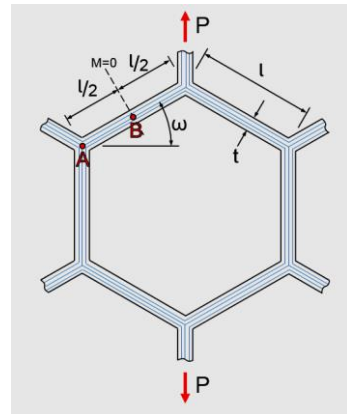
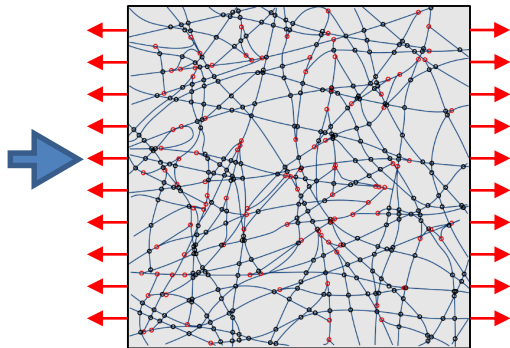
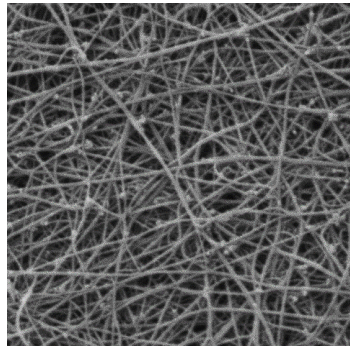
Modulus below that of CNTs because...

1. Network structure causes **foam-like** network deformation.
2. Bundles are **rope-like**.

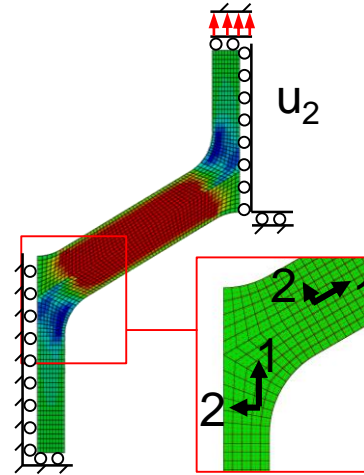
Micromechanical Model for direct-spun mat



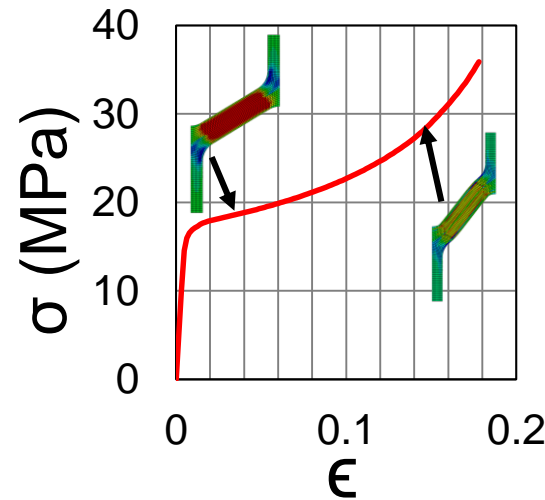
CNT Bundles are **Anisotropic**:
 $E_{11} = 680 \text{ GPa}$
 $G_{12} = G_{23} = 9.5 \text{ GPa}$
 $E_{22} = E_{33} = 50 \text{ GPa}$
 $\nu_{12} = \nu_{13} = \nu_{23} = 0.3$



Approximate network with a **periodic honeycomb** unit cell

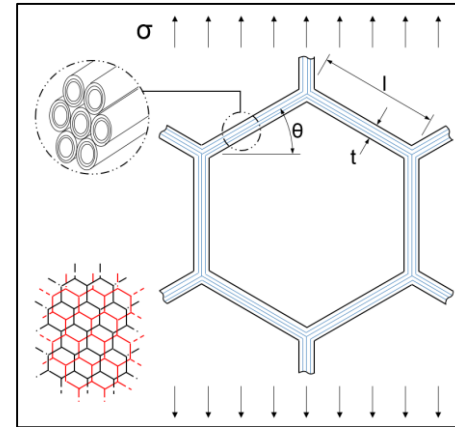
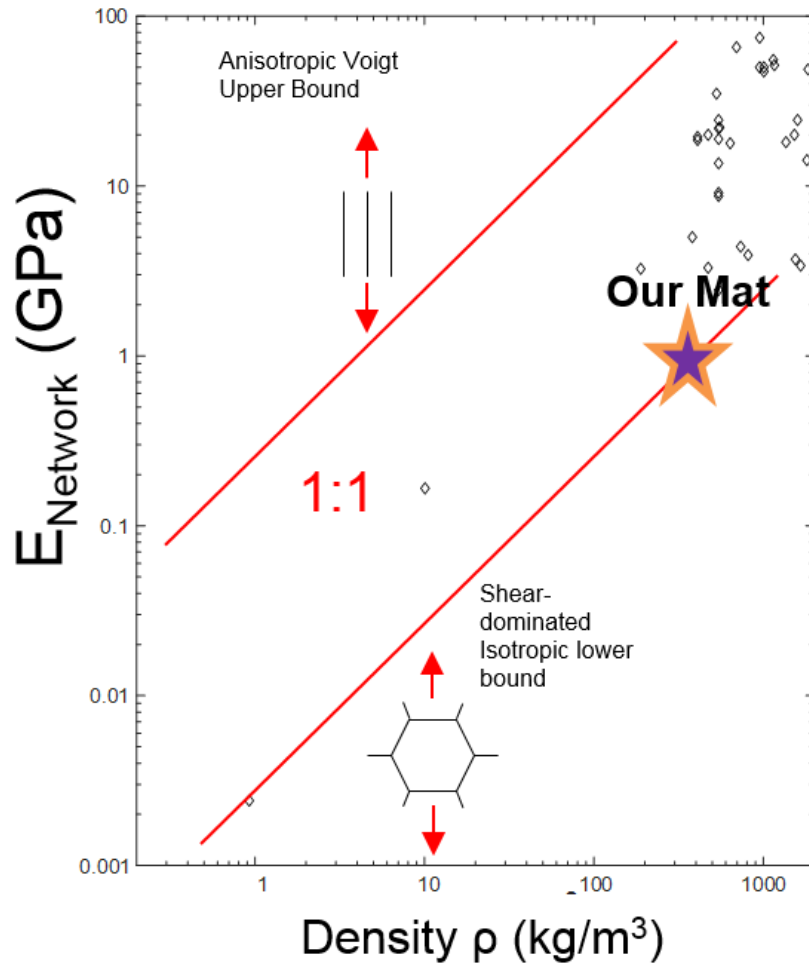


Macroscopic yield dictated by the **shear strength** of CNT bundles.



Network reorientation causes **hardening**

Routes for Improvement...



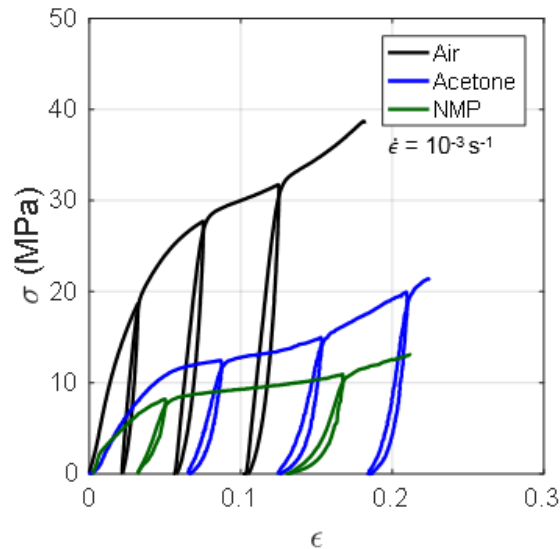
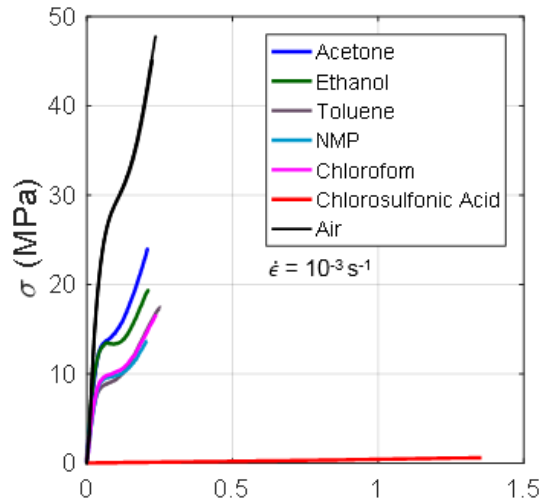
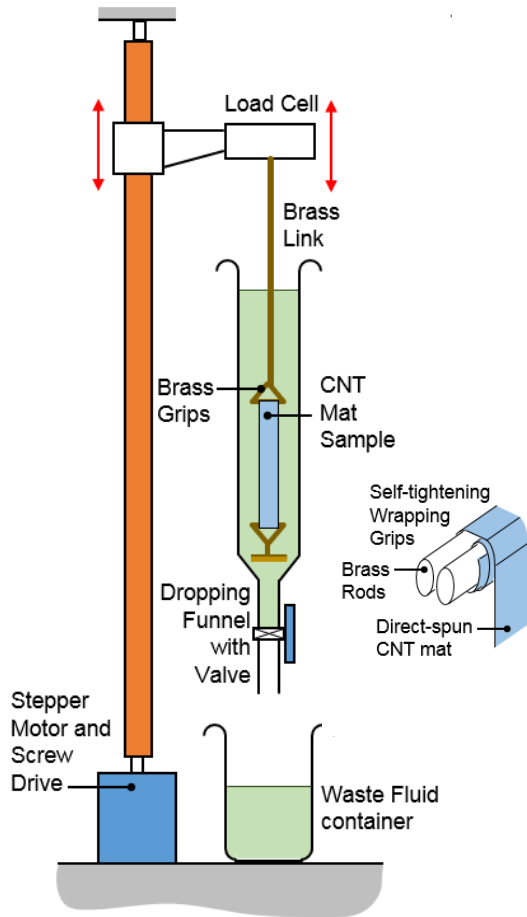
for $G_B \ll E_B$:

$$E_{Network} = \bar{\rho} \cdot \frac{\frac{2}{3}\kappa G(1+\sin \theta)^2}{\cos^2 \theta}$$

Therefore, improvement in mechanical properties can come from **ALIGNMENT of CNT BUNDLE MICROSTRUCTURE**



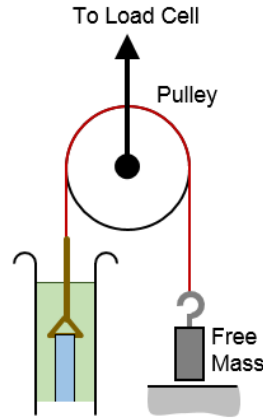
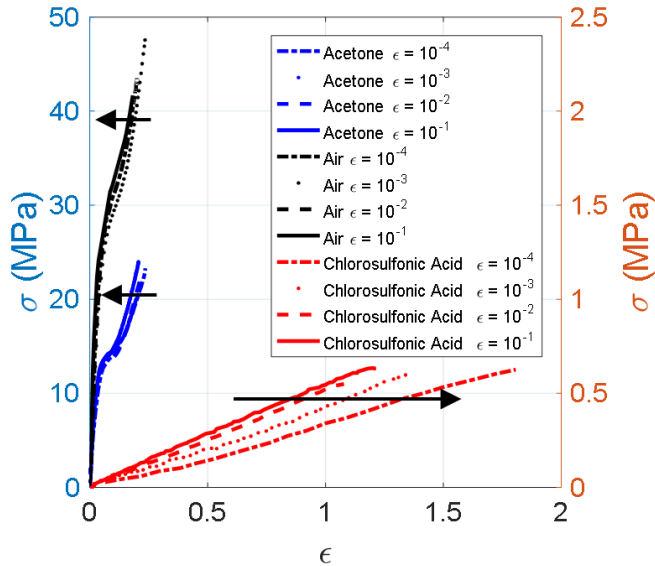
Response in Fluids



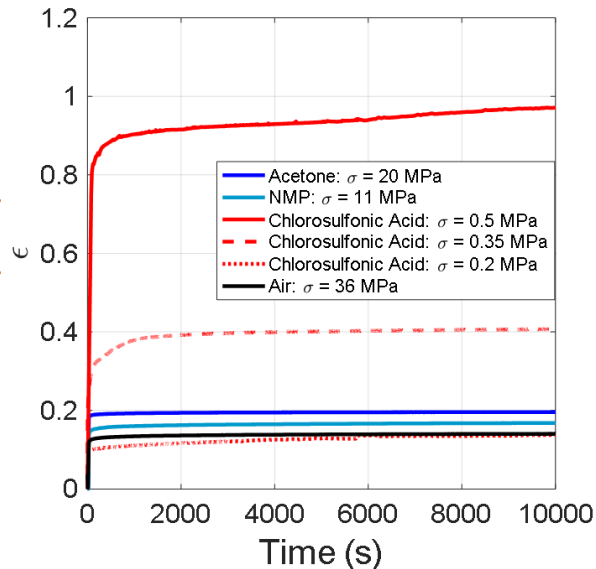
- Chlorosulfonic acid lowers σ and E by over an order of magnitude.
- ϵ increases to ~ 1.4 at same rate.



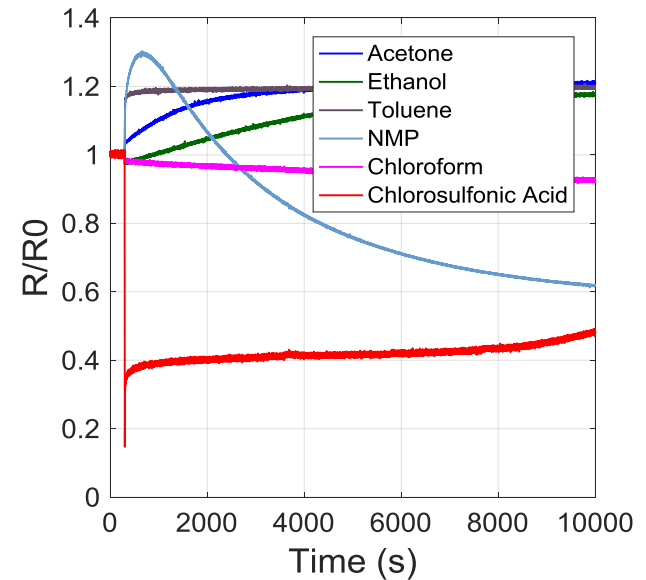
Response in Fluids



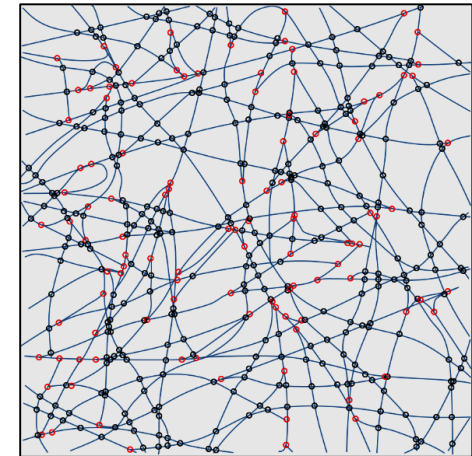
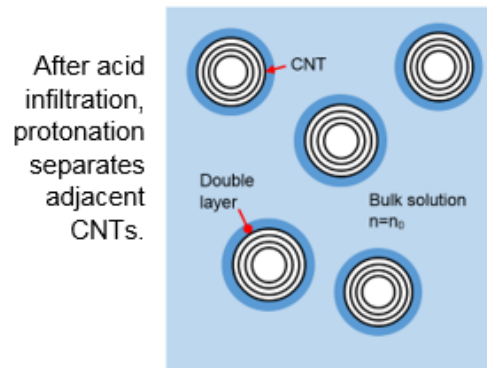
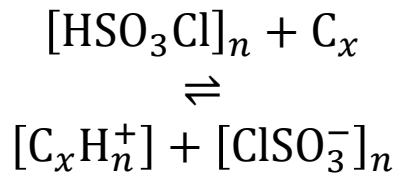
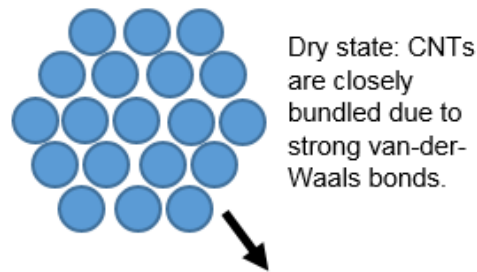
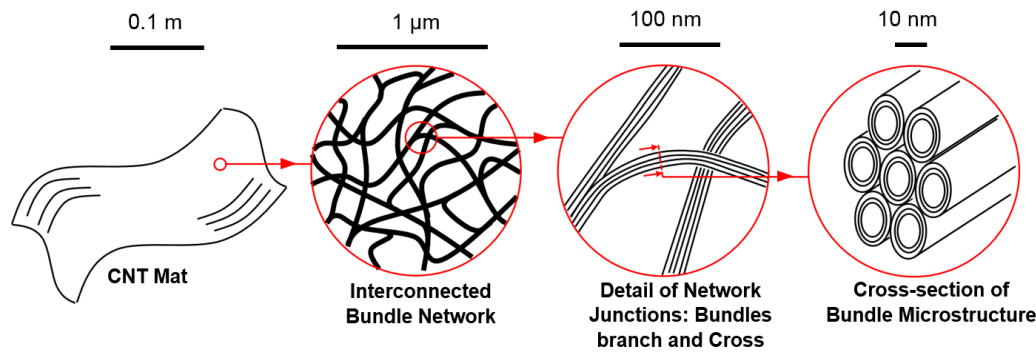
Electrical resistance also affected by fluid immersion... but **mechanical behaviour is time invariant.**



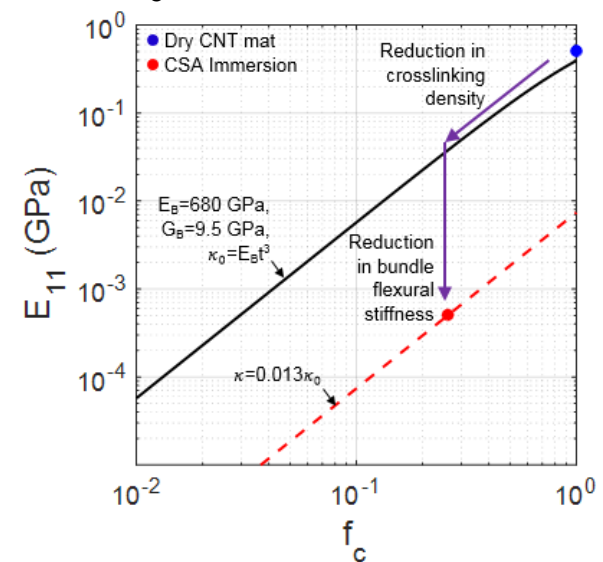
Immersion in **chlorosulfonic acid** results in **creep** at low stresses.



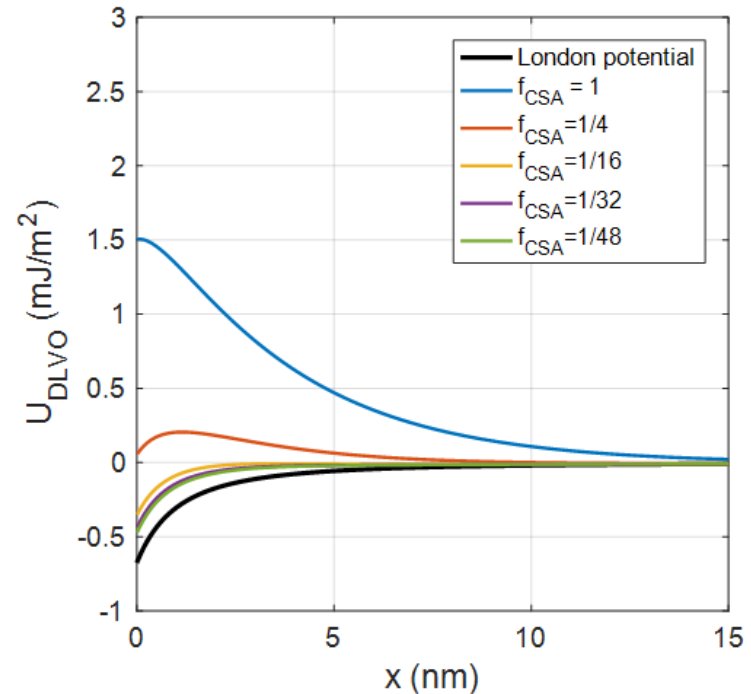
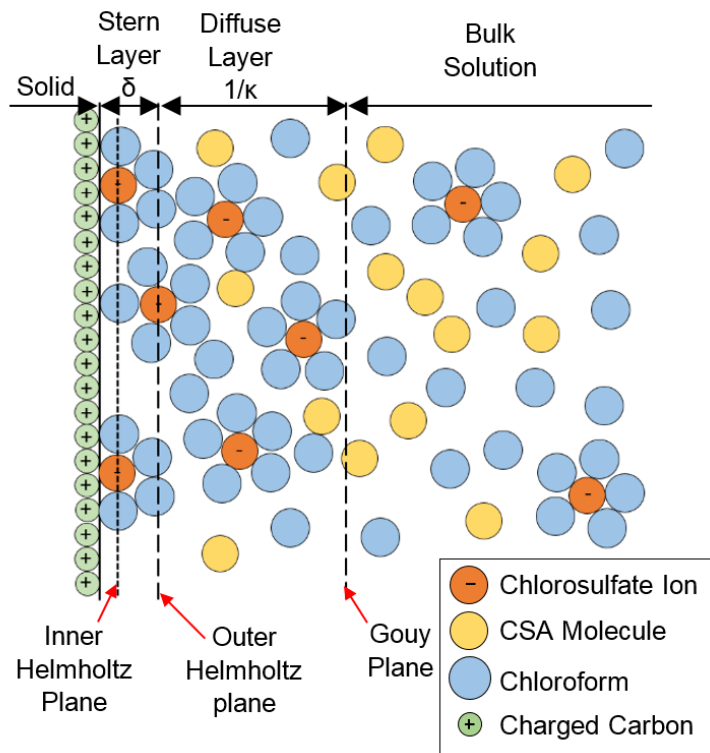
Debundling/debonding upon CSA Immersion



$f_c \sim 0.26$



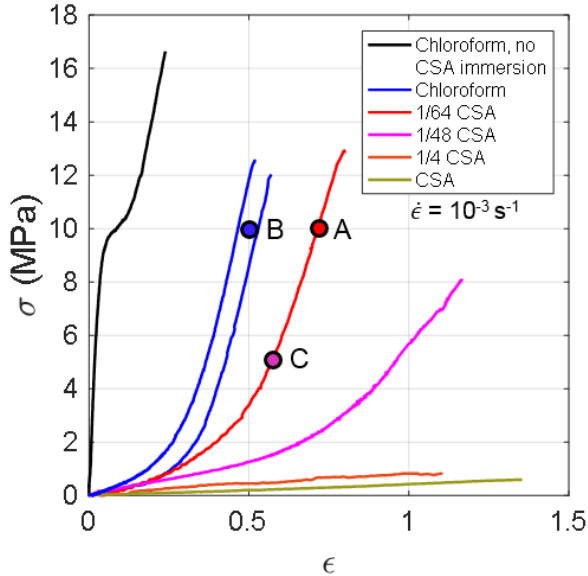
Debundling/debonding upon CSA Immersion



The presence of **adsorbed ions** at the CNT wall, and in the **solution** screen the positive charge upon the CNT walls, and overcome the **van-der-Waals** attraction.

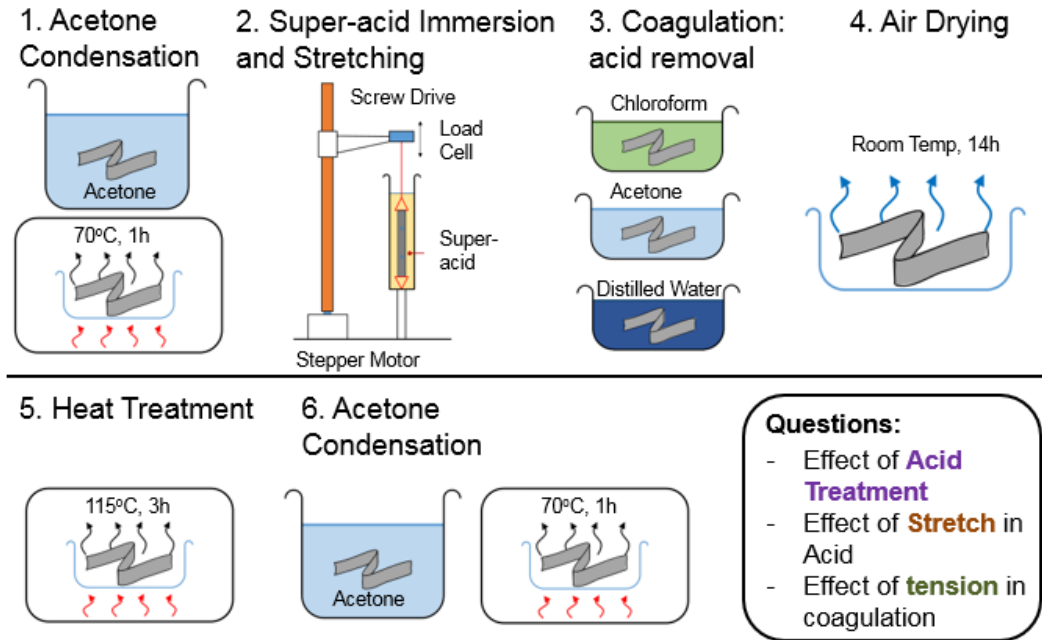


Fluid Processing in superacid solutions



Ductility and drawing stress controlled by the concentration of a superacid solution.

Drawing process to enhance alignment

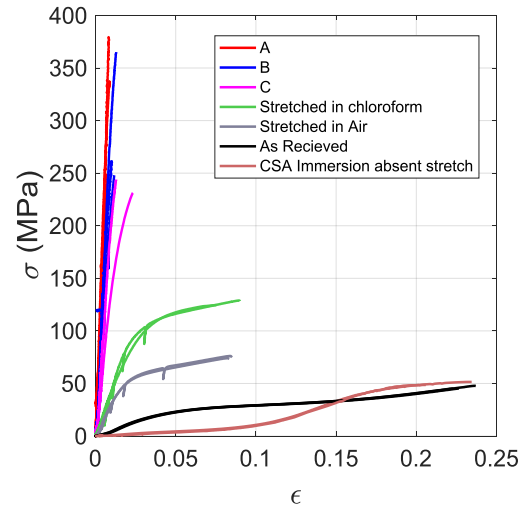
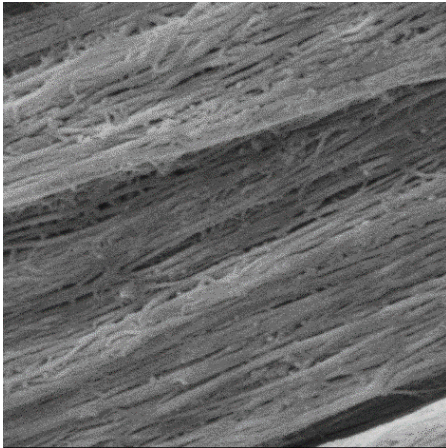
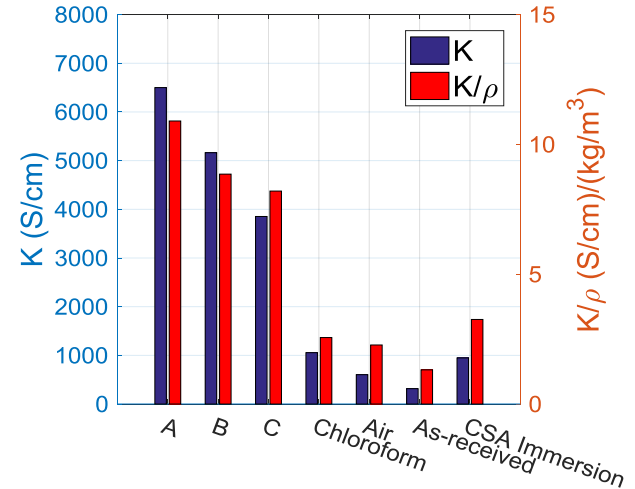
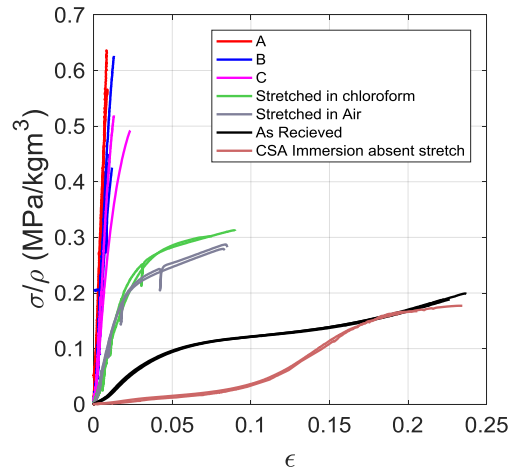
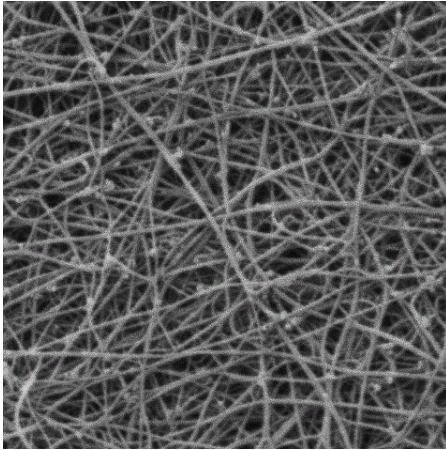


Questions:

- Effect of **Acid Treatment**
- Effect of **Stretch** in Acid
- Effect of **tension** in coagulation



Properties of drawn fibres



- All properties improved significantly.
- Change in ultimate specific strength and conductivity a factor of 3.
- Larger change in stiffness due to switch away from bending.

1 μm



Summary

- The **properties** of direct-spun carbon nanotube materials (and CNT materials in general) vary across a **wide range of density**.
- The stiffness and strength of direct-spun mats is **reduced** by the **CNT bundle network** of **low nodal connectivity**, and by the rope-like structure of the CNT bundles.
- Mechanical and electrical properties of direct-spun CNT mats are **enhanced** by tensile drawing in different fluids, particularly in **superacids**.

