

Seeing what the eye cannot see

Using DIC and in-situ SEM to study the micromechanics of pearlitic steel

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Full field deformation and strain measurements with a focus on
optical techniques

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Introduction

Full-field measurements

- ▶ Topic of this seminar: Full field deformation and strain measurements
- ▶ How can we use full-field measurements to learn more about materials?
 - ▶ Several examples during this seminar
 - ▶ Qualitative
 - ▶ Quantitative
- ▶ Present study: use data from DIC analysis of in-situ SEM micrographs of a pearlitic steel

Introduction

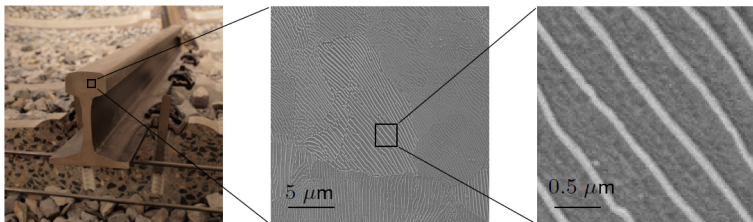
Engineering applications



- ▶ Applications: suspension bridges, cable cars, railway applications, ...
- ▶ Severe loads during manufacturing and/or operation
- ▶ Loading \Leftrightarrow Microstructure \Leftrightarrow Mechanical behavior
[WETSCHER, TORIBIO, ...]

Introduction

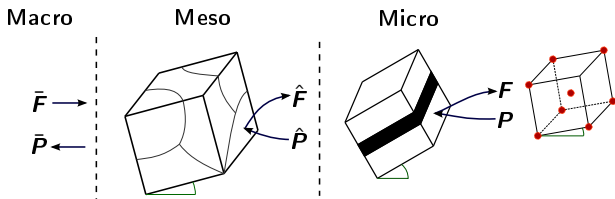
Length scales



- ▶ Macroscale
 - ▶ Homogeneous appearance
 - ▶ Initially isotropic
- ▶ Mesoscale
 - ▶ Colonies (constant lamella orientation)
 - ▶ Local anisotropy
 - ▶ Severe loads: alignment of cementite lamellae
- ▶ Microscale: cementite lamellae (brittle) embedded in ferrite (ductile)
- ▶ Next: how to include such knowledge in a constitutive model?

Multiscale modeling

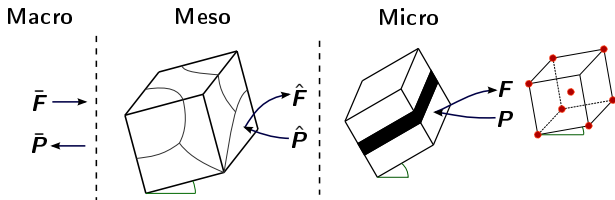
Preliminaries



- ▶ Main idea: replace macroscopic constitutive model with a Representative Volume Element (RVE) [FEYEL, MIEHE, GEERS, WRIGGERS, ZODHI ...]
- ▶ Enables known subscale features and mechanisms to be included in the model
- ▶ Model calibration: macroscopic data and/or subscale data
- ▶ Computational cost
- ▶ Virtual experiments

Multiscale modeling

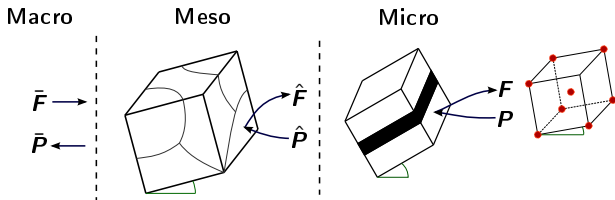
Mesoscale RVE



- ▶ Local macroscopic response obtained from homogenization of mesoscale RVE
- ▶ Aggregate of colonies
- ▶ Meso-RVE needs to include a sufficient number of colonies to be representative
- ▶ Mechanical behaviour of individual colonies
- ▶ Interactions between colonies
- ▶ Local mesoscopic response obtained from homogenization of microscale RVE

Multiscale modeling

Microscale RVE



- ▶ Models the mechanical behaviour of the two phases: cementite and ferrite
- ▶ Takes into account how they interact with each other
- ▶ Constitutive framework needed to tie things together: crystal plasticity
- ▶ Note: model parameters are defined on microscale

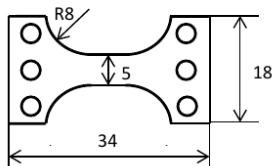
Present study

Objectives and outline

- ▶ Collaboration with:
 - ▶ Johan Ahlström (Chalmers)
 - ▶ Magnus Colliander (Chalmers/GKN)
 - ▶ Ru Peng (Linköping University)
- ▶ Analyze in-situ SEM micrographs of pearlite using DIC
- ▶ Increased understanding of microscale mechanisms
- ▶ Investigate possibilities for using microscale data for model calibration
- ▶ Microstructure used as a natural speckle
 - ▶ How to handle regions ill-posed for DIC-procedure?
- ▶ No distortion correction!

Present study

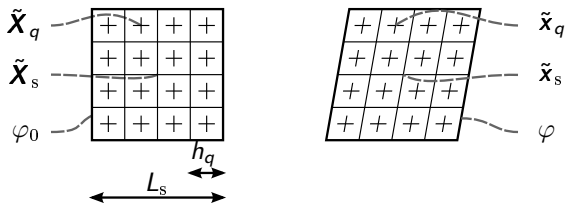
Experimental setup



- ▶ Test material: ASTM 1080 carbon steel
- ▶ Test equipment: Hitachi SU-70 FEG- SEM with a Gatan microtest tensile stage
- ▶ Experiments carried out in cooperation with Linköping University
- ▶ Macroscopic data: applied force
- ▶ Subscale data: SEM micrographs at varying global load levels
- ▶ Micrographs to be processed using DIC

Present study

DIC - preliminaries



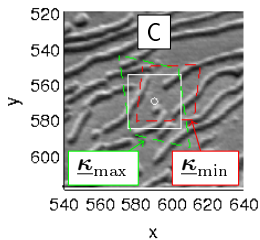
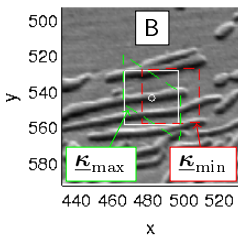
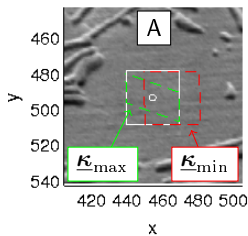
- Definition of correlation function [SUTTON, REU, HILD, ROUX, RANSON, PETERS, SCHREIER, VENDROUX...]

$$c(\tilde{\mathbf{X}}, \underline{\kappa}, t) = \frac{1}{N_q} \sum_{q=1}^{N_q} [\gamma_q - \gamma_{0,q}]^2 \quad (1)$$

- Subset kinematics

$$\tilde{\mathbf{x}}_q(\underline{\kappa}) = \begin{cases} \tilde{\mathbf{X}}_q + \tilde{\mathbf{u}}_s & k = 0 \\ \tilde{\mathbf{X}}_q + \tilde{\mathbf{u}}_s + \tilde{\mathbf{H}}_s \cdot [\tilde{\mathbf{X}}_s - \tilde{\mathbf{X}}_q] & k = 1 \\ \dots & \dots \end{cases} \quad (2)$$

Identification of high quality subsets

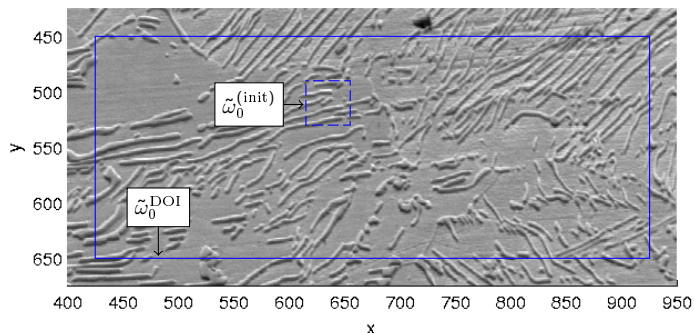


- ▶ Synthetic speckles vs. natural speckle
- ▶ Traceability indicators: initial standard deviation and eigenvalues of the Hessian of the autocorrelation function

$$\underline{\mathbf{K}}_a \cdot \underline{\kappa}^{(I)} = \lambda^{(I)} \underline{\kappa}^{(I)} \text{ where } \begin{cases} \underline{\mathbf{K}}(\tilde{\mathbf{X}}, \underline{\kappa}, t) = \frac{\partial^2 c}{\partial \underline{\kappa} \otimes \partial \underline{\kappa}} \\ \underline{\mathbf{K}}_a(\tilde{\mathbf{X}}) \equiv \underline{\mathbf{K}}(\tilde{\mathbf{X}}, \underline{\kappa} = \mathbf{0}, t = 0) \end{cases}$$

Identification of high quality subsets

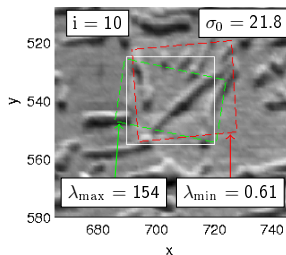
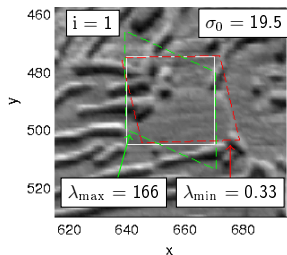
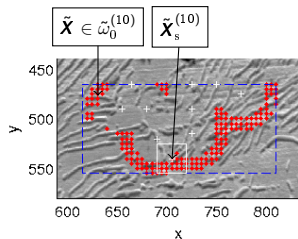
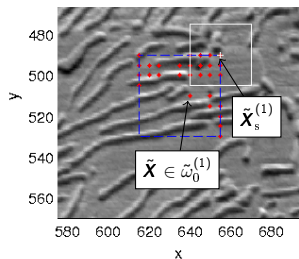
Algorithm



- ▶ Dense background grid of subsets
- ▶ Would like to cover the DOI with subsets which have a well-defined correlation functions

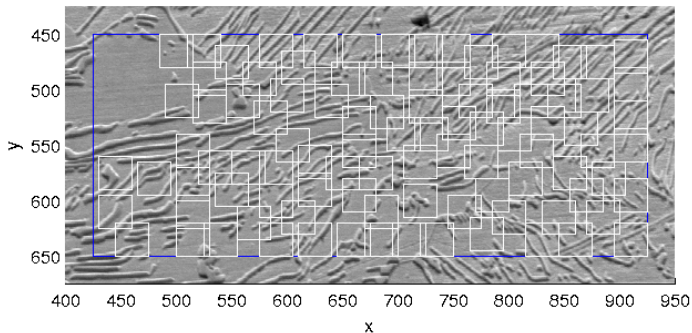
Identification of high quality subsets

Iterative procedure



Identification of high quality subsets

Identified subsets



- ▶ The domain is covered with subsets that fulfill the traceability criteria
- ▶ Domains with limited information quality are avoided

Error estimation

- ▶ A known motion is applied to an initial image

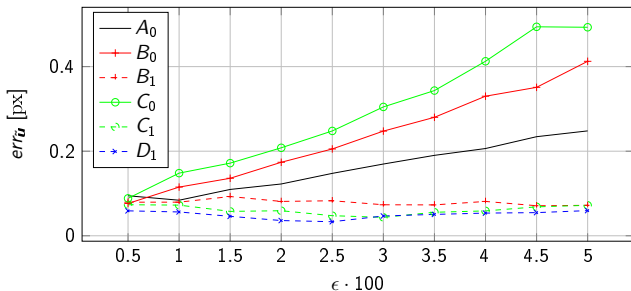
$$[\tilde{\mathbf{x}} - \tilde{\mathbf{x}}_R] = [\tilde{\mathbf{H}} + \mathbf{I}] \cdot [\tilde{\mathbf{X}} - \tilde{\mathbf{X}}_R] \tilde{\mathbf{u}} = \tilde{\mathbf{H}}[\tilde{\mathbf{X}} - \tilde{\mathbf{X}}_R]$$

- ▶ Uniaxial elongation in the horizontal direction is considered
- ▶ Both displacements and the displacement gradient is known in the entire image domain
- ▶ A number of setups are evaluated

k	0	0	0	1	1	1
L_s	10.5	20.5	30.5	20.5	30.5	40.5
Label	A_0	B_0	C_0	B_1	C_1	D_1
N_s	2087	1755	755	1915	838	557

Error estimation

Local displacements



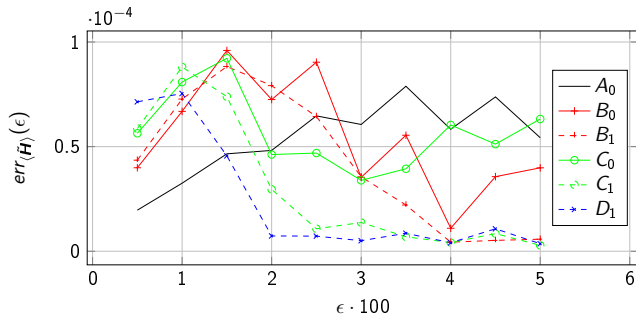
- ▶ The local displacement error is introduced:

$$err_{\tilde{\mathbf{u}}} = err_{\tilde{\mathbf{u}}}(\epsilon) = \max_i (|\tilde{\mathbf{u}}(\tilde{\mathbf{X}}_s^{(i)}, \epsilon) - \tilde{\mathbf{u}}_s^{(i)}|)$$

- ▶ Using first order kinematics gives better results
- ▶ Subset size has little impact ($k=1$)

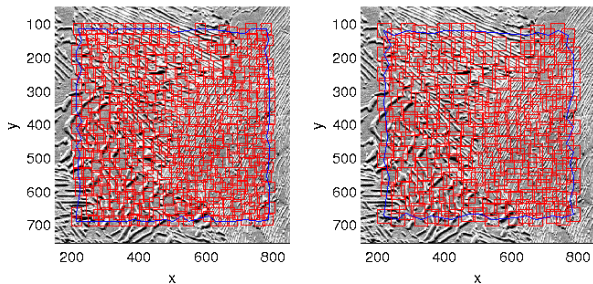
Error estimation

Average displacement gradient



- ▶ Using first order kinematics improves the results
- ▶ Using a larger subset improves the results
- ▶ Note: Constant displacement gradient in domain

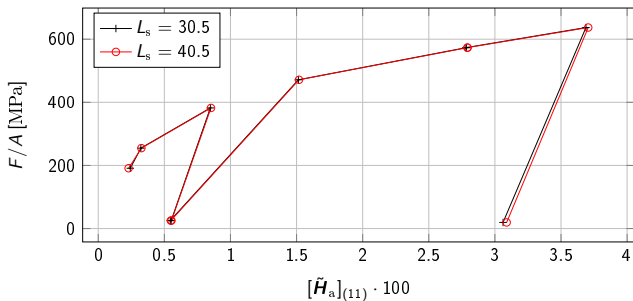
DIC analysis of in-situ SEM micrographs



- ▶ Two subset sizes are considered: $L_s = 30.5$ [px] and $L_s = 40.5$ [px]
- ▶ The procedure for identification of high quality subsets is used

DIC analysis of in-situ SEM micrographs

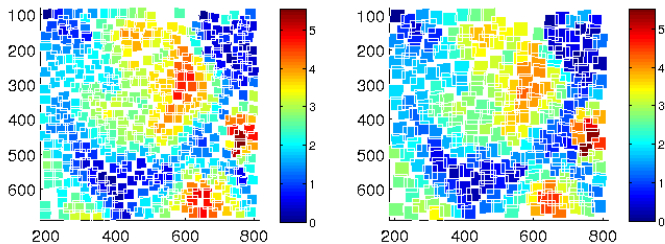
Identified strain history



- ▶ Results are rather insensitive to the chosen subset size (robustness)
- ▶ Note: reloading path not resolved




DIC analysis of in-situ SEM micrographs

Fluctuation field



- ▶ Impact of material morphology and crystallographic orientations can be visualized using the fluctuation fields
- ▶ The two analyses give qualitatively similar results
- ▶ Grain like domains can be identified (compare with EBSD maps)

Concluding remarks

- ▶ In-situ obtained SEM images have been analyzed using Digital Image Correlation
- ▶ The microstructure of the material is used as a natural speckle
- ▶ A method for identification of high-quality subsets within a DOI is proposed
- ▶ The implementation of the DIC framework is validated using synthetically deformed images
- ▶ The method is robust (insensitive to the chosen subset size) when it comes to identifying displacements and the average displacement gradient
- ▶ Future work
 - ▶ Influence from sub-surface morphology (tomography+DVC)?
 - ▶ Compensate for SEM image distortion
 - ▶ FE-DIC
 - ▶ Parameter value identification (FEMU, VFM, ...)
- ▶ Acknowledgement:
 - ▶ Linköping University
 - ▶ Swedish Research Council
 - ▶ The Areas of Advance in Materials Science and Transport,   

Concluding remarks

Questions?